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Synthesis, characterisation and application of TiO₂–zeolite nanocomposites for the advanced treatment of industrial dye wastewater



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ABSTRACT

Photocatalysis usually involves the utilisation of nano-sized semiconductor photocatalysts owing to their higher specific surface area and surface reaction rate. However, the key challenges in the utilisation of nano-sized photocatalysts for advanced treatment of industrial dye wastewater are to enhance the post-separation and recovery of spent photocatalysts to prevent them from diffusing into the environment. Thus, the main aim of this study was to synthesize a functional-form of titanium dioxide (TiO₂)-zeolite nanocomposite through the modified two-step sol-gel method for enhanced application and separation after advanced industrial dye wastewater treatment. The synthesized TiO₂-zeolite nanocomposite was characterised using field-emission scanning electron microscopy (FE-SEM), energy dispersive X-ray (EDX) analysis, Fourier-transformed in-frared spectroscopy (FTIR), particle size distribution analysis and Brunauer–Emmett–Teller (BET) specific surface area and porosity analysis. Subsequently, the photoactivity of synthesized TiO₂-zeolite nanocomposite shows a high apparent pseudo-first order reaction rate constant of 0.0419 min⁻¹ at lower dye concentration. This showed that the synthesized TiO₂-zeolite nanocomposite follows a more adsorption-oriented photocatalytic degradation of water pollutants, which is useful for removing trace and untreated dye compounds in the advanced industrial dye wastewater treatment stage.

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1. Introduction

In recent years, rapid industrialisation has extended the use of synthetic dyes in various industrial applications in order to meet the escalating demands on consumer products to sustain the national economic growth. While this is a favourable and positive shift in an endeavour to shape the short-term or long-term economic outlook of a country; the environmental dimension of effective handling and management of produced industrial dye wastewater was often being neglected. Industrial dye wastewater is highly refractory, as it is usually made up of high salt and organic contents that render its low biodegradability in the natural environment [1,2]. In order to

safeguard and prevent the bioaccumulation of synthetic dye compounds in the natural environment, direct discharge of industrial dye wastewater effluent to environmental waterways or even local wastewater treatment plants (WWTPs) should be made illicit [3]. Instead, the industrial dye wastewater should be retained on-site for further purification before discharging to the local WWTPs [3]. This is to ensure that most of the toxic and hazardous parent azo-aromatic synthetic dyes and their intermediate compounds are being treated to avoid this point source pollution from diffusing into the natural ecosystem.

To date, various concerted efforts have been made to retain the industrial dye wastewater for on-site treatments such as by using the natural pond and lagoon treatments; physico-chemical treatments such as sedimentation, flocculation, coagulation and adsorption; biological treatments such as aerobic and anaerobic digestion or even high-rate algal metabolism and chemical treatments such as Fenton, photo-Fenton and ozonation [4,5]. However, the treatment of industrial dye wastewater is not straightforward as synthetic

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dye compounds are usually having very good water solubility. Together with the highly stable azo-aromatic structures of synthetic dyes, these made the industrial dye wastewater even more difficult to be degraded naturally or via bio-assisted degradation processes. In addition, the breakdown of synthetic dye compounds would release by-products such as benzidine, naphthalene and other aromatic compounds that are toxic, carcinogenic and mutagenic not only to marine living organisms but also affecting human beings through the food chain. For instance, Lupica [6] reported the toxicological effects of synthetic dyes could have on the shape and size of red blood cells of fishes. In addition, it was also reported that the untreated synthetic dyes can cause the reduction of reproductive organs of rat up to 44%, decrease in total protein concentration by 70% and cholesterol was depleted up to 91% [6].

Photocatalysis is an emerging branch of advanced oxidation technologies used for water and wastewater treatment that has been widely documented for its ability to degrade indiscriminately almost all water pollutants [4]. Nano-sized semiconductor photocatalysts are usually being used owing to their higher specific surface area and thus, giving yields to a higher surface reaction rate during the degradation of water pollutants. The utilisation of nano-sized photocatalysts in the advanced treatment of industrial dye wastewater is, however, remained highly challengeable owing to the problems associated with post-separation and recovery of spent photocatalysts [7]. Even with the additional post-separation and recovery stage; the photo-efficiency of catalysts, recovery efficiency of photocatalysts and unit treatment cost of photocatalysis are the major factors that inhibit the large scale application of photocatalytic water treatment technology [8]. Thus, the key challenge in the utilisation of nano-sized photocatalysts for advanced treatment of industrial dye wastewater is to provide a bespoke solution by immobilising the nano-sized semiconductor photocatalysts on larger immobiliser substrates to alleviate the post-separation and recovery efficiency. Different immobiliser substrates such as magnetite core [9], activated carbon [10] and clays [7,11] have been used to enhance their post-separation and recovery from bulk water.

In comparison, natural zeolites are abundant, easily available and inexpensive as the immobiliser substrate to synthesize functional nanocomposites with semiconductor metal oxides [12]. More interestingly, zeolites are having crystalline aluminosilicates with different cavity structures and high ion exchange capacity, tectosilicates with microporous channels, pore spaces, molecular sieving, adsorption and catalysis capacity [13,14]. Recently, the titanium dioxide (TiO₂)-zeolite nanocomposite has been studied for the removal of humic acid from drinking water source [12]. Thus, the main aim of this study was to design a functional-form of TiO₂-zeolite nanocomposite by varying the acid concentrations, zeolite loadings and annealing temperatures used in the modified two-step sol-gel synthesis process for subsequent application in the advanced treatment of industrial dye wastewater. In this instance, nano-sized TiO₂ crystals were synthesized and immobilised on submicron sized zeolite particles to enable the functionality of "nanoparticles" while presenting a microstructure for ease of post-separation and recovery after industrial dye wastewater treatment. Following this, further characterisations on the physicochemical properties and photoactivity of TiO₂–zeolite nanocomposite were carried out using field-emission scanning electron microscopy (FE-SEM), energy dispersive X-ray (EDX) analysis, Fourier-transformed infrared spectroscopy (FTIR), particle size distribution analysis and Brunauer-Emmett-Teller (BET) specific surface area and porosity analysis. Finally, the effects of catalysts loading, annealing temperature and initial dye concentration on the photodegradation kinetics of surrogate model Reactive Black 5 dye compound were investigated. The UV-vis absorbance scans from 400 to 700 nm were also being carried out to monitor the cleavage and disappearance of characteristic π – π system linked to the azo bonds and the possible appearance of other reaction by-products. With this study,

it is anticipated that photocatalytic water treatment technology utilising TiO₂-zeolite nanocomposite can provide a technically feasible and cost effective solution for advanced industrial dye wastewater treatment.

2. Materials and methods

2.1. Materials

Titanium (IV) butoxide (97% gravimetric, Sigma–Aldrich) and absolute ethanol (Kollin Chemicals) and zeolites (particle size < 45 μ m, Sigma–Aldrich, Product code: 96096) were used as received. Nitric acid (Kollin Chemicals), sodium hydroxide (Kollin Chemicals) and Reactive Black 5 dye ($C_{26}H_{21}N_5Na_4O_{19}S_6$, 55% Sigma–Aldrich) were prepared to the final desired concentration via the addition of deionised water.

2.2. Preparation of TiO₂-zeolite nanocomposites

The modified two-step sol-gel synthesis method for TiO₂-zeolite nanocomposites was in accordance to Chong et al. [7] in exception that kaolinite was replaced by zeolite particles. Briefly, 25 mL of titanium (IV) butoxide was mixed with 30 mL of absolute ethanol under vigorous magnetic stirring. The mixture was denoted as mixture A. Then 60 mL of diluted nitric acid was added drop wise into mixture A. During the mixing process, the mixture A was stirred continuously until the dispersion first became milky white and eventually transparent homogeneous sol with no visible precipitation. The stirring was continued for another 30 min to ensure the homogeneity in transparent homogenous sol formed. Following this, the zeolite particles were made into suspension via the addition of 100 mL deionised water and the zeolite suspension was set in flask immersed in water bath with temperature set at 37 °C. The prepared transparent sol was added drop wise into the zeolite suspension and the mixture was continuously stirred for 4 h. Subsequently, the final mixture was cooled to room temperature and aged for 13-16 h. After the aging process, the mixture was filtered and washed repeatedly for three times with deionised water and then the filtrate was dried at 65-70 °C for 2-4 h. Finally, the filtrate containing TiO₂-zeolite nanocomposites was annealed at different temperatures before it was used as the photocatalysts for the treatment of synthetic industrial dye wastewater.

2.3. Characterisation of TiO_2 -zeolite nanocomposites

Field-emission scanning electron microscopy images were obtained using the Hitachi SU8010 electron microscope at an accelerating voltage of 5 kV. Similarly, the energy dispersive X-ray analysis was carried out using the same electron unit after the samples were coated with platinum for ease of analysis of percentages of atomic element and element weight, respectively.

Brunauer–Emmett–Teller specific surface area of the TiO₂–zeolite nanocomposites was determined by using adsorption isotherms obtained from the Micromeritics BET ASAP 2020 (surface area and porosity analyser) at 77 \pm 0.5 K in liquid nitrogen and by using the BET equation. Sample vessels were degassed at high temperatures overnight before being analysed for SSA, pore volume and pore size of the TiO₂–zeolite nanocomposites.

The functional groups on TiO₂-zeolite nanocomposites were analysed using Thermo Scientific FTIR Nicolet iS10 while particle size distribution were determined using Malvern Mastersizer 3000.

2.4. Photocatalytic experiments on aqueous Reactive Black 5 solution

The photoactivity of synthesized TiO_2 -zeolite nanocomposites was tested on the degradation of a surrogate indicator, Reactive Black 5 dye, which is commonly found in industrial dye wastewater. The



Fig. 1. Effects of different HNO₃ concentration on the textural and gelation characteristics of TiO₂ sol-gel formed.

reaction solution was placed in a beaker on a magnetic stirrer under a UV illumination of approximately 300 μ W/cm². Air ventilation in the reactor box was controlled using a vacuum pump with the ventilation rate set at 2 L/min. Sampling was done during the reaction using a micropipette.

During the initiation of photocatalytic experiment, 100 mL of the reaction solution was placed in the beaker with certain photocatalysts loading. The reaction solution was then magnetically stirred for 30 min to ensure homogeneous mixing before the UV illumination was turned on. Subsequently, the samples were collected every 30 min up until 180 min of reaction time. The collected samples were centrifuged (ScanSpeed Mini, Labogene, Denmark) at 12,000 rpm for 15 min before the supernatants were being separated and filtered for analysis. A triplicate of the reaction solution was sampled where the absorption spectra were scanned using a UV–vis spectrophotometer (Genesys 10uv, Thermo Electron Corporation). The monochromatic maximum wavelength of 597 nm from the UV–vis scanning was used, to determine the concentration of Reactive Black 5 dye in the samples.

In order to evaluate for the photoactivity of TiO_2 -zeolite nanocomposites, the raw data on concentration-time was fitted with the Langmuir–Hinshelwood model as given in Eq. (1). The reason for selecting this kinetic model to represent the heterogeneous photocatalysis reaction was due to that the model dye compound is thought to adsorb on the catalysts surface before being degraded and finally, the products will be desorbed from the catalysts surface [4]. Depending on the dye concentration, Chong et al. [15] found that the Langmuir–Hinshelwood model could be further simplified when the *KC*-value is less than 1 to the simplified pseudo-first order kinetic model as given in Eq. (2).

$$r = \frac{dC}{dt} = \frac{kKC}{1+KC} \tag{1}$$

$$\ln\left(\frac{C_0}{C}\right) = kKt = k_{app}t\tag{2}$$

where r is the reaction rate, k is the reaction rate constant, K is the dynamic Langmuir adsorption constant, C is the dye concentration and k_{app} is the apparent pseudo-first order reaction rate constant.

3. Results and discussion

3.1. Effects of nitric acid concentration

The synthesis of TiO₂ sol–gel constitutes the first step in the modified two-step sol–gel synthesis method as previously reported by Chong et al. [7]. The Ti-precursor solution was first hydrolysed in a hydrolysis reaction followed by condensation. Through this synthesis method, it is possible to control the homogeneity of TiO_2 sol-gel as well as tune the microstructures of eventual TiO_2 crystallites formed. In order to yield nanosized TiO_2 crystallites, the control on the extent of hydrolysis reaction is essential. In this study, the hydrolysis of Ti-precursor was carried out in two stages, where a partially hydrolysed state was first targeted by adding absolute ethanol to the Ti-precursor. Chong and Jin [8] explained that this partially hydrolysed state is to provide non-hydrolysable ligands of Ti for better TiO_2 cluster-to-cluster growth during condensation, as well as to provide better molecular homogeneity. Due to the low functionality range of ethanol in the hydrolysis of Ti-precursor, the true extent of hydrolysis reaction was controlled by the second HNO₃ acid-catalysed hydrolysis reaction.

Fig. 1 shows the effects of varying HNO₃ concentration during the acid-catalysed hydrolysis reaction step on the textural and gelation characteristics of TiO₂ sol-gel formed. The range of studied HNO₃ concentration was between 0.05 M and 1 M. In this instance, the variation in the HNO₃ concentration resulted in TiO₂ sol-gel of different fluidity and precipitation. Previously, Chong et al. [7] reported that the optimum HNO₃ concentration lies in the range of 0.25–0.30 M without any temperature and humidity controls during the synthesis experiments. From Fig. 1, however, it was observed that there are some precipitations in lumps still occur at HNO₃ concentration below 0.35 M. Precipitation was observed at low acid concentration due to a very high gelation rate that resulted in precipitates in the TiO₂ polymeric gel network. For the synthesis of TiO₂-zeolite nanocomposites, precipitates are not favourable for the immobilisation of TiO₂ crystallites onto the external surface of zeolite due to its low degree of fluidity (i.e. non-flowing hardened TiO₂ sol-gel). When the acid concentration of 0.30–0.35 M was used, it was found the hardening of gel occurred almost instantaneously after the addition and stirring with HNO₃. No precipitation was observed for HNO₃ concentration above 0.35 M where instead, a clear hard gel was formed at acid concentration range of 0.35–0.40 M. As for high HNO₃ concentration above 0.5 M, no gelation occurred even after long ageing time and Ti-sol presented a very high fluidity characteristic. It was understood that high acid concentration would slow down the condensation rate and thus, affecting the formation of TiO₂ polymeric gel.

When the acid concentration of 0.38 M was used, the gel formed hardened at a lower rate allowing the Ti-sol to be added drop-wise into the zeolite suspension in later stage. Thus, the optimum HNO₃ concentration found in this study which showed a good gelation behaviour with acceptable fluidity level and no precipitation occurred

at 0.38 M. This was not in agreement with the previous finding reported by Chong et al. [7] and this was deduced to be owing to higher temperature and local humidity level in the present experiment settings. A systematic study should be carried out in order to understand the effects of temperature and humidity on the eventual textural and gelation characteristics of TiO₂ sol-gel formed. Apart from getting the right textural and gelation characteristics of TiO₂ sol-gel, it was also known that the optimum condition will also affect the TiO₂ crystallites size formed. In this study, since the volume of acid used was kept constant while the molarities were varied, the predominant synthesis parameter was related to the pH. Su et al. [16] also reported that the TiO₂ crystallite particles in aqueous solution possess surface charges that are highly dependent on pH. When the surface charges of TiO₂ particles are high, strong repulsive forces exert among the particles making them less likely to form aggregates and thus, resulting in more stable Ti-sol with smaller particle sizes. It was also reported that large TiO₂ particles sizes are formed between pH 5 and pH 8 where white precipitates were formed. Clear Ti-sol formed at low pH was reported to form nano-sized TiO₂ particles at pH less than pH 3. This is consistent with the results obtained in this study where, the optimum TiO₂ sol-gel condition chosen was clear and contained no precipitates.

3.2. Effects of zeolite loading

In the modified two-step sol-gel synthesis method for TiO_2 -zeolite nanocomposites, the latter part was to coat TiO_2 sol-gel onto the external surfaces of zeolite before subjected to controlled annealing treatment. During the optimisation of TiO_2 sol-gel condition, the acid-catalysed hydrolysis reaction was manipulated to produce sol-gel with optimal fluidity to enable drop wise addition into the zeolite suspension. This was done to enable the proper and more uniform coating of positively-charged TiO_2 sol-gel dispersed onto the negatively-charged aluminosilicate framework of zeolite. It is theorised that the TiO_2 sol-gel will only be immobilised on the external surfaces of zeolite without being dispersed into its pores or cavities.

Fig. 2(a) and (b) shows the FE-SEM images showing the changes in surface morphology of zeolite, both before and after TiO₂ loading. From Fig. 2(a) and (b), it can be observed that a distinct change in surface morphology occurred when TiO₂ was heterogeneously immobilised on the zeolite surface. In order to further investigate the effects of zeolite loadings on TiO₂ dispersion, different mass loadings of zeolite in suspensions of 5%, 10% and 15% w/v was mixed with a constant volume of Ti-sol prepared from the first synthesis procedure. Fig. 2(c)-(e) shows the FE-SEM images of TiO₂-zeolite nanocomposites synthesized under three different mass loadings of zeolite at 5% (w/v), 10% (w/v) and 15% (w/v), respectively. From Fig. 2(c)–(e), the analysis of three different mass loadings of zeolite used revealed that the TiO₂ particles were not evenly distributed over the zeolite surface and many sections of the zeolite surface still remained bare and uncovered by TiO_2 particles. When a low zeolite loading of 5% (w/v) was mixed with constant volume of Ti-sol, the TiO₂ layer on zeolite surface was found to be thick and layered coating (Fig. 2(c)). This was possibly due to the aggregation of TiO₂ particles during the immobilisation process and is believed to cause a reduction in the effective surface area of TiO₂ particles that had immobilised on the zeolite surface. It was however, at higher mass loadings of zeolite, the TiO₂ layer on the zeolite surface showed a consistent and layered coating and this coating appeared to become thinner when the zeolite loading was increased (*i.e.* 15% w/v). This was attributed to the higher amount of crystal nucleation sites available for TiO₂ binding when higher zeolite loading was used. Finally, an energy dispersive X-ray analysis was carried out to validate the presence and composition of Ti in the prepared TiO₂-zeolite nanocomposites. Table 1 shows the quantified elemental composition of prepared TiO2-zeolite nanocomposites from EDX analysis and this confirmed that the layered coating formed on the zeolite surface was indeed due to the immobilisation

of TiO₂ particles. Since the amount of Ti added remains unchanged throughout the synthesis experiments, both the weight and atomic percentages of Ti in the TiO₂-zeolite nanocomposites are measured (using EDX) to be approximately 23.83% and 10.02%, respectively.

3.3. Characterisation of TiO₂-zeolite nanocomposites

3.3.1. BET adsorption isotherm analysis

By analysing the adsorption isotherms of nitrogen on the mesoporous zeolite surface at an increasing relative pressure, the BET surface area, pore size and volume were determined. Fig. 3 shows the adsorption and desorption isotherms of TiO₂-zeolite nanocomposites for zeolite loading of 5% w/v and annealing treatment at 400 °C. From Fig. 3, it can be observed that the adsorption and desorption isotherms are not the same for a specified region of relative pressures. This phenomenon is known as a hysteresis loop and is commonly exhibited in mesoporous adsorbents such as zeolite particles used in this study. With the adsorption and desorption isotherm characteristics, the observed isotherms can be classified as the Type IV isotherm in accordance to the IUPAC isotherm classifications [17].

Table 2 shows the effects of zeolite loading and annealing temperature on the BET surface area, pore size and pore volume distribution. It was observed that an increase in zeolite loading from 5% to 10% (w/v) in suspension is accompanied by an increase in BET surface area, as the mesoporous zeolite structure is porous. At the zeolite loading of 5% (w/v), it was found through FE-SEM imaging (Fig. 2(f)) that TiO₂ crystallites were dispersed more frequently on the zeolite surface due to lesser nucleation sites. When the pore volume was examined for mass loadings of zeolite at 5% and 10% (w/v), two potential hypotheses can be portrayed. Firstly, the immobilisation of TiO₂ crystallites on the external surface of zeolite is causing blockage of zeolite pores and resulted in an attenuation of pore volume from $0.223379 \text{ cm}^3/\text{g}$ to 0.201215 cm³/g. Secondly, the immobilised TiO₂ layer on zeolite could form a microporous structure that could increase the pore volume of TiO₂-zeolite nanocomposites. It was however, from the BET analysis that the zeolite pore sizes were ranged between 2 and 3 nm while the TiO₂ crystallites were measured from 10 to 100 nm. Thus, the difference in zeolites pore size and TiO₂ crystallites size confirmed that the second hypothesis is valid whereas a microporous TiO₂ layer was formed on the external zeolite surface.

When the effect of annealing temperatures was studied between 300 °C and 600 °C, the TiO₂-zeolite nanocomposites were found to exhibit a wide rangeability of behaviour in terms of the BET surface area, pore size and volume. By ensuring the mass loading of zeolite was constant at 5% (w/v), the increase in annealing temperature from 300 °C to 400 °C was observed to have negligible effect on the BET surface area but a slight increase in both the pore size and volume. However, when the annealing temperature was increased to 600 °C, an abrupt reduction in the BET surface area from 279.5122 \pm 1.1593 m²/g to $38.1123 \pm 1.2562 \text{ m}^2/\text{g}$ was observed. Vimonses et al. [18] reported that there exists a small endothermic peak at 600 °C in zeolite that corresponds to the dehydroxylation process in zeolite. This endothermic peak was the potential reason for the abrupt reduction in BET surface area as observed when the annealing temperature was increased from 400 °C to 600 °C. Simultaneously, an increase in both the pore size and volume was observed and this is in close agreement with the observed endothermic peak in zeolite at 600 °C.

3.3.2. Fourier-transformed infrared spectroscopy (FTIR)

The functional groups that present in TiO_2 -zeolite nanocomposites were studied using the FTIR spectroscopy. Fig. 4 shows the comparison of FTIR spectra of pure zeolite and TiO_2 -zeolite nanocomposites. From Fig. 4, it can be observed that the samples of pure zeolite and TiO_2 -zeolite nanocomposites annealed at 500 °C and 600 °C exhibited similar FTIR spectra characteristics of zeolite. This indicated



Monash 2.0kV 4.2mm x7.00k SE(U)





Fig. 2. FE-SEM images showing changes in the surface morphology of zeolite (a) before TiO₂ loading; (b) after TiO₂ loading; (c) with 5% w/v zeolite; (d) with 10% w/v zeolite; (e) with 15% w/v zeolite and (f) measured TiO₂ crystallites size on zeolite.

that zeolite is high suitable to act as TiO₂ immobiliser, as the structure is not affected by the high annealing temperatures used.

The broad band occurring at 3286 cm⁻¹ in both annealed TiO_2 zeolite nanocomposites samples was linked to the -OH stretching and bending vibrations of silanol groups (Si-OH), which was formed due to the interaction between -Si groups in zeolite and water molecules. While the small band at 1619 cm⁻¹ is a characteristic of the bending vibrations of H-O-H bonds in water molecules associated with zeolite particles. Following these, the sharp intense peak observed at 958 cm⁻¹ was due to the Si-O-Si or Al-O-Al non-symmetric stretching vibrations in the tetrahedral SiO₄ and AlO₄ structures that form the zeolite framework and thus, explaining its band intensity being



Fig. 3. Adsorption and desorption isotherms of TiO₂-zeolite nanocomposites for zeolite loading of 5% w/v and annealing treatment at 400 °C.

 Table 1

 Elemental composition of prepared TiO₂-zeolite nanocomposites from EDX analysis.

| Element | Weight% | Atomic% |
|---------|---------|---------|
| 0 | 65.1181 | 81.9005 |
| Al | 5.34493 | 3.98998 |
| Si | 5.7023 | 4.08635 |
| Ti | 23.8347 | 10.0231 |
| Total | 100 | 100 |

the highest. Previous study by Smirnov and Graaf [19] and Damin et al. [20] reported that the insertion of Ti into the zeolite structure would result in the Ti–O–Si bonds at the infrared band of 960 cm⁻¹. It was also reported that the band intensity is proportional to the amount of Ti present in the zeolite framework. However, the FTIR spectra at the intense band of 958 cm⁻¹ that represents the Si and Al tetrahedral bonding was found to overshadow the potential Ti band formed. This observation indicated that the amount of Ti that presents in the TiO₂– zeolite nanocomposites could be minimal and again proven that the microporous TiO₂ crystallites layer is surface-bounded without any interstitial binding into the zeolite structure.

3.3.3. Particle size distribution

Previously, the TiO₂ crystallites on nanocomposites were measured as shown in Fig. 2(f). From the average TiO₂ crystallites size estimation over sampling through FE-SEM image, the size range was found between 10 nm and 80 nm. While Fig. 5 shows the particle size distribution of TiO₂-zeolite nanocomposites measured using the laser light diffraction method. From Fig. 5, it was observed that the size rangeability of TiO₂-zeolite nanocomposites varied from 0.4 to $200 \,\mu$ m with the majority of the nanocomposites particles in the size range of 5–10 µm. The larger particle sizes measured were due to the aggregation of zeolite particles and this is shown in the FE-SEM imaging in Fig. 6. In Fig. 6, it was observed that the zeolite aggregates formed were composed of cubes arranged in a random and disorganised manner. As for the particle size distribution below 5 μ m, this was found to be owing to the non-aggregated single cubes of zeolite. It was also observed that when the zeolite cubes aggregate, they will form cavities and channels in between the cubes. In this instance, the TiO₂ crystallites were seen to bind more frequently to the cavities and



Fig. 4. Comparison of the FTIR spectra of pure zeolite and $\ensuremath{\text{TiO}_2}\xspace$ -zeolite nanocomposites.

channels, as compared to the flat surfaces of zeolite particles. With such a nanocomposites structure of TiO_2 and zeolite, it is anticipated that they could be exhibiting the "nano-scale" effects while a submicron sized microstructure for ease of separation after wastewater treatment.

3.4. Photoactivity evaluation of TiO₂-zeolite nanocomposites

3.4.1. Effect of catalyst loading

Catalyst loading is an important parameter in photocatalytic water treatment processes [21]. The optimum catalyst loading enables the maximum photoactivity while preventing the unnecessary excess use of applied catalysts. Fig. 7 shows the plot of pseudo-first order rate constant of model Reactive Black 5 dye against TiO₂-zeolite nanocomposites loadings used. Control experiment showed that in the absence of UV irradiation, the photocatalytic degradation of Reactive Black 5 dye was very low to negligible. In the other control experiment with the absence of TiO₂-zeolite nanocomposites photocatalyst, the direct UV photolysis of Reactive Black 5 was still evident with a measured pseudo first-order rate constant of 0.0035 min⁻¹ (data not shown). From Fig. 7, it was found that the optimum TiO₂-zeolite nanocomposites loading required is 0.3 g/L that yields the pseudo-first order rate constant of 0.0102 min⁻¹. Primarily the increase in photoactivity when the TiO₂-zeolite nanocomposites loading was increased from 0.1 g/L to 0.3 g/L was attributed to the higher number of active sites and more reactive radicals available for surface reaction. However, the reduction in photoactivity of TiO2-zeolite nanocomposites was observed at higher catalysts loadings owing to the increasing cloudiness in the reaction solution that prevents penetration from UV illumination. Mahadwad et al. [22] also explained that the increase in catalysts concentration will result in the deactivation of activated molecules due to collisions with ground state molecules. Thus, the optimum TiO₂-zeolite nanocomposites for the photocatalytic degradation of 10 ppm model Reactive Black 5 dye in aqueous solution were 0.3 g/L.

3.4.2. Effect of annealing temperature

Fig. 8 shows the effects of annealing temperature on TiO_2 -zeolite nanocomposites on its photoactivity in degrading the model Reactive Black 5 dye in aqueous solution. It was interesting to note that

Table 2

Effects of different zeolite loading and annealing temperature on the BET surface area, pore size and pore volume.

| Zeolite loading | Annealing temperature (°C) | BET surface area (m²/g) | Pore size (nm) | Pore volume (cm ³ /g) |
|-----------------|----------------------------|--|----------------|----------------------------------|
| 5% zeolite | 400 | $\begin{array}{r} 279.4826 \pm 1.6744 \\ 324.0927 \pm 1.7259 \end{array}$ | 2.87983 | 0.201215 |
| 10% zeolite | 400 | | 2.75698 | 0.223379 |
| 5% zeolite | 300 | $\begin{array}{r} 279.5122 \ \pm \ 1.1593 \\ 38.1123 \ \pm \ 1.2562 \end{array}$ | 2.79353 | 0.195206 |
| 5% zeolite | 600 | | 4.32126 | 0.831245 |







Fig. 6. FE-SEM images showing (a) large aggregates of zeolite and/or TiO₂-zeolite nanocomposites; (b) non-aggregated single cube of zeolite and/or TiO₂-zeolite nanocomposites.



Fig. 7. Plot of pseudo-first order rate constants of model Reactive Black 5 dye against TiO_2 -zeolite nanocomposites loadings used.

the TiO₂-zeolite nanocomposites sample annealed at 600 °C showed higher photoactivity than those annealed at 300 °C, albeit lower BET specific surface area for sample annealed at 600 °C. The apparent pseudo-first order rate constants at both 300 °C and 600 °C were 0.0056 min⁻¹ and 0.0102 min⁻¹, respectively. The higher photoactiv-



Fig. 8. Effects of annealing temperature on the photocatalytic degradation of model Reactive Black 5 dye in aqueous solution. Initial dye concentration: 10 ppm; pH 5 and TiO₂-zeolite nanocomposites loading: 0.3 g/L.

ity in TiO_2 -zeolite nanocomposites sample annealed at 600 °C could be attributed to the formation of higher pore size and volume as discussed in Section 3.3.1. Another physicochemical property that contributes to the higher photoactivity is due to the presence of photoactive TiO₂ phases, of whether pure anatase or rutile TiO₂ or a



Fig. 9. Comparison plot between TiO₂-zeolite nanocomposites and commercial TiO₂ particles in apparent pseudo-first order rate constants against initial dye concentration. Inset: Percentage degradation of dye against UV illumination time at initial dye concentration: 1 ppm; pH 5 and catalyst loading of 0.3 g/L.

mixture of the two photoactive phases. From our previous study, we found that the pure anatase TiO_2 photoactive phase dominates where the mixed anatase/rutile phase only appears when annealed at 650 °C [7].

3.4.3. Effect of initial dye concentration

Fig. 9 shows the comparison plot of apparent pseudo-first order rate constants between TiO₂-zeolite nanocomposites and commercial TiO₂ particles on photocatalytic degradation of different initial dye concentrations. From Fig. 9, it was found that the TiO_2 -zeolite nanocomposites exhibited a higher apparent pseudo-first order rate constant of 0.0419 min⁻¹ than the commercial TiO₂ particles (*i.e.* 0.0297 min^{-1}) at a low dye concentration of 1 ppm. This was comparatively higher than the apparent pseudo-first order rate constants obtained at higher dye concentrations for both TiO₂-zeolite nanocomposites and commercial TiO₂ particles. At low dye concentration, dye molecules are dynamically adsorbed, reacted and desorbed from the external surface of TiO₂-zeolite nanocomposites. As the dye concentration increases, cumulative amount of dye molecules are adsorbed onto the TiO₂-zeolite nanocomposites that reduces the penetration of UV illumination and rate of surface reaction, respectively. Simultaneously, the absorption of UV illumination by the surface-adsorbed dye molecules also attenuates the photon transmissivity to the catalyst surface. The inset in Fig. 9 shows the comparison plot in photocatalytic degradation kinetics of Reactive Black 5 dye at 1 ppm for both TiO₂-zeolite nanocomposites and commercial TiO₂ particles. From the kinetics plot, it was observed that a complete degradation of Reactive Black 5 dye was achieved in 95 min and 120 min using the TiO₂-zeolite nanocomposites and commercial TiO₂ particles, respectively. While at higher initial dye concentration of 10 ppm, it was observed that the commercial TiO₂ particles (*i.e.* 0.0128 min⁻¹) performed better than the TiO₂-zeolite nanocomposites (*i.e.* 0.0102 min⁻¹). Fig. 10 shows the attenuation in UV-vis absorbance peak at λ_{max} value of 597 nm at 30 min interval, as a result of the cleavage and disappearance of characteristic π system linked by the azo bonds ($-N \equiv N-$). Based on these findings, it can be concluded the TiO₂-zeolite nanocomposites exhibited more adsorption oriented photocatalytic degradation that could be useful to remove



Fig. 10. Absorbance of Reactive Black 5 reaction solution samples taken at 30 min intervals from 400 to 700 nm (10 ppm initial dye concentration, pH 5, 25 °C, 0.3 g/L catalyst).

trace water pollutants in the advanced water/wastewater treatment stages.

4. Conclusion

In conclusion, the modified two-step sol-gel method was successfully adopted to synthesize TiO_2 -zeolite nanocomposite with surface covered TiO_2 crystallites size of 10–100 nm and zeolite immobilisers size of 5–10 μ m. This enables the functionality of "nano-sized effect" for TiO_2 while presenting sub-micron zeolite microstructure platforms to allow for the ease of post-separation and recovery after advanced industrial dye wastewater treatment. During the first sol-gel synthesis step, the optimum acid concentration was found to be 0.35–0.40 M HNO₃ that yielded a homogeneous, clear and hard TiO_2 sol-gel with no precipitation formed. Subsequently, the TiO_2 sol-gel was coated onto different mass loadings of zeolite and the physical structural existence of TiO_2 -zeolite nanocomposite was successfully characterised and proven using FE-SEM and EDX analysis. FE-SEM images showed that a consistent layer of TiO_2 crystallites was present on the zeolite surface, while the EDX analysis confirmed the Ti composition in the synthesized TiO₂-zeolite nanocomposite. The different zeolite loadings were seen to increase the BET surface area by providing more surface nucleation sites for the growth of TiO_2 crystallites. Surface characterisation using BET analysis showed that the TiO₂-zeolite nanocomposite exhibited the adsorption isotherm characteristic similar to mesoporous zeolite, which indicates that the high specific surface area and adsorption capacity characteristics were preserved. FTIR analysis conducted showed that the amount of Ti that presents in the TiO₂-zeolite nanocomposite was minimal and also proved that the microporous TiO₂ crystallites layer formed was surface-bounded without any interstitial binding into the zeolite structure. Photocatalytic degradation of model Reactive Black 5 dye showed that the highest apparent pseudo-first order reaction rate constant of 0.0419 min⁻¹ was achieved at the optimum TiO₂zeolite nanocomposite loading of 0.3 g/L and initial dye concentration of 1 ppm. More interestingly, it was found that the $\rm TiO_2-zeolite$ nanocomposite annealed at 600 °C showed higher photoactivity than those annealed at 300 °C, albeit lower BET specific surface area for sample annealed at 600 °C. In this instance, the higher photoactivity in TiO₂-zeolite nanocomposites sample annealed at 600 °C could be attributed to the formation of higher pore size and volume in the TiO₂-zeolite nanocomposite. Finally, a comparison in photoactivity between the TiO₂-zeolite nanocomposite and commercial TiO₂ particles revealed that synthesized nanocomposite exhibited a higher apparent pseudo-first order rate constant and degradation kinetics at lower dye concentration. This could be an important implication that the TiO₂-zeolite nanocomposite followed more adsorption-oriented photocatalytic degradation that could be useful to remove trace water pollutants in the advanced water/wastewater treatment stage.

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